

# Melting Experiments on Fe-O-H and Fe-H: Evidence for Eutectic Melting in Fe-FeH and Implications for Hydrogen in the Core

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## Key Points:

- We examined the liquidus phase relations and solid/liquid partitioning in the Fe-O±H and Fe-H systems at ~40 GPa and ~150 GPa.
- Eutectic melting between Fe and FeH is indicated from the phase relations and melting temperature in Fe-O-H and Fe-H.
- The outer core may include 2.9–5.2 wt% O, 0.03–0.32 wt% H, 0–3.4 wt% Si, and 1.7 wt% S.

**Abstract** We examined liquidus phase relations in Fe-O±H at ~40 and ~150 GPa, and in Fe-H at 45 GPa. While it has been speculated that Fe and FeH form continuous solid solution to core pressures, our experiment on Fe-H showed that FeH<sub>0.20</sub> forms with the hcp structure, different from fcc for stoichiometric FeH, and melts at temperature lower than that for FeH, suggesting eutectic melting between Fe and FeH. It is consistent with the liquidus phase diagram in Fe-O-H, which implies the Fe-FeH binary eutectic liquid composition of FeH<sub>0.42</sub> at ~40 GPa. We estimated the outer core liquid composition to be Fe + 2.9–5.2% O + 0.03–0.32% H + 0–3.4% Si + 1.7% S by weight, based on the liquidus phase relations, solid-liquid partitioning, and outer/inner core densities and velocities, indicating that O and either H or Si are important core light elements.

**Plain Language Summary** We have investigated the melting phase relations in the Fe-O±H and Fe-H systems at high pressures in a laser-heated diamond-anvil cell. The solid-liquid partition coefficient of H was also determined. While it is known that Fe and stoichiometric FeH form continuous solid solution at least to ~20 GPa, our experiments on the Fe-O-H ternary and Fe-H binary systems performed at ~40 GPa consistently suggested eutectic melting between Fe and FeH with eutectic liquid composition of FeH<sub>0.42</sub>. We explored the possible range of the Earth's liquid core

composition, which should be 1) within the liquidus field of Fe to crystallize the dense inner core, 2) compatible with seismological observations of the outer core, and 3) in chemically equilibrium with the inner core solid that explains the observed density and velocities. The results indicate that the outer core is rich in O and either H or Si, supporting the delivery of a large amount of water to the Earth found in recent planet formation theories and its sequestration into the metallic core that is also inferred from metal-silicate partitioning data.

## 1. Introduction

The Earth's core should contain more than one light impurity elements (see recent reviews by Li & Fei, 2014 and Hirose et al., 2021). Both O and H can be important core light elements because the density and sound velocity observed in the outer core are compatible with the presence of O (Badro et al., 2014) and H (Umemoto & Hirose, 2015, 2020). Recent planet formation models suggested that 10 to 100 times ocean mass of water may have been brought to the Earth during its accretion (e.g., Raymond et al., 2007; Walsh et al., 2011). The chemical reaction of solar-nebula-type H-rich proto-atmosphere with a magma ocean could have been another source of water (Ikoma & Genda, 2006; Olson & Sharp, 2019). It is likely that most of H<sub>2</sub>O was once dissolved into a magma ocean, and H and O were incorporated into core-forming metals during their segregation from silicate (Tagawa et al., 2021; Li et al., 2020; Yuan & Steinle-Neumann, 2020).

Liquidus phase relations of Fe alloy systems, in particular the liquidus field of Fe (a compositional range of liquids that first crystallize Fe), constrain the outer core composition (e.g., Yokoo et al., 2019; Hasegawa et al., 2021; Sakai & Hirose, 2021). When O and H are two major core light elements, the liquid core should not crystallize FeO nor FeH but Fe at the inner core boundary (ICB), otherwise the denser solid inner core is not formed. In order to understand the liquidus phase relations in the Fe-O-H system, the knowledge on the relevant binary systems is helpful. Previous high-pressure experiments demonstrated that O concentration in the Fe-FeO eutectic liquid increases with increasing pressure (Morard et al., 2017; Oka et al., 2019). The recent study by Oka and others based on experiments to 204 GPa estimated 15 wt% O in the Fe-FeO binary eutectic liquid at ICB. On the other hand, earlier high-pressure melting experiments on the Fe( $\pm$ Ni)-H system were limited to 20 GPa (Sakamaki et al., 2009; Imai, 2013; Shibazaki et al., 2014) because of a technical difficulty in dealing with hydrogen. Fukai (1992) speculated continuous solid solution between Fe and

69 stoichiometric FeH to megabar pressure, but it has been verified only to ~20 GPa (Imai,  
70 2013; Shibazaki et al., 2014).

71 In this study, we have carried out melting experiments on the Fe-O±H and Fe-H systems  
72 at ~40 GPa and ~150 GPa. Our results on Fe-O-H suggest that Fe-FeH is an eutectic  
73 system, which contradicts the phase diagram supposed by Fukai (1992) but is supported  
74 by our experiment on Fe-H alloy. Our findings of 1) eutectic melting between Fe and  
75 FeH, 2) liquidus phase relations in the Fe-O-H system, and 3) the solid-liquid  
76 partitioning of H help to constrain concentrations of O, H, and other light elements in  
77 the outer core.

## 78 **2. Results**

79 We have examined the melting phase relations in the Fe-O±H and Fe-H systems on the  
80 basis of high-pressure and -temperature ( $P$ - $T$ ) experiments using laser-heated  
81 diamond-anvil cell (DAC) techniques, combined with in-situ synchrotron X-ray  
82 diffraction (XRD) measurements and ex-situ textural and compositional  
83 characterizations using electron microprobes (see Experimental Methods in the  
84 Supporting Information).

### 85 **2.1. Melting in Fe-O±H**

86 A total of six separate melting experiments were carried out in the Fe-O±H system at  
87 ~40 GPa and ~150 GPa (runs #A1–A6 in [Table S1](#)). For all runs, we performed  
88 synchrotron XRD measurements and subsequent microprobe observations on recovered  
89 samples. In run #A4, while the intense peaks from hcp Fe were present before heating,  
90 they were almost lost and alternatively a diffuse signal characteristic of liquid was  
91 observed around 12 degrees of two-theta angle upon heating to 2370 K ([Figure 1a](#)).  
92 Upon quenching temperature, the peaks of fcc FeH<sub>x</sub> ( $x = 0.31$ ) appeared from the liquid  
93 portion. The XRD pattern obtained at ~12 μm away from the center of a laser-heated  
94 spot included strong peaks from rhombohedrally-distorted B1 FeO and hcp FeH<sub>0.24</sub>.

95 This sample was recovered from a DAC, and in its cross section, there was a chemically  
96 homogeneous area with a non-stoichiometric composition at the center, which  
97 represents a quenched liquid ([Figures 2a, b](#)). The earlier DAC experiments by Hirose et  
98 al. (2019) and Tagawa et al. (2021) demonstrated that the H content in FeH<sub>x</sub> crystals  
99 formed from liquid upon quenching temperature closely represents that of Fe-H liquid.  
100 We therefore consider that Fe-O-H liquids in the present experiments were fully

101 quenched into a mixture of solid  $\text{FeH}_x$  and  $\text{FeO}$ . The proportion of  $\text{FeO}$  was obtained  
 102 from the EPMA analysis of O concentration in the quenched liquid. The remaining  
 103 phase proportion of  $\text{FeH}_x$  and its H concentration  $x$  that was estimated from its lattice  
 104 volume in XRD patterns give 0.49 wt% H in the Fe-O-H liquid (Table S1). Next to the  
 105 liquid pool, electron microprobe images showed a couple of single-phase layers of Fe  
 106 and  $\text{FeO}$  (Figures 2a, b). According to the XRD observations, the Fe layer had been  
 107 solid  $\text{FeH}_{0.24}$  (Fe + 0.43 wt% H) before it lost hydrogen during decompression (Figure  
 108 1a). The partition coefficient of H between solid Fe and liquid,  $D_H$  (solid/liquid), was  
 109 0.89 on weight basis. The liquid was in direct contact with the  $\text{FeO}$  layer or in some  
 110 place with the  $\text{FeH}_{0.24}$  layer, indicating that both  $\text{FeO}$  and  $\text{FeH}_{0.24}$  are the liquidus phases  
 111 of the Fe-O-H liquid formed in run #A4. The outside of these liquidus phase layers was  
 112 not melted. Runs #A1–A3 were carried out around 40 GPa in a similar manner with  
 113 changing the Fe/ $\text{H}_2\text{O}$  ratio in the starting materials (Table S1). These liquids included  
 114 0.1–0.4 wt% C except 1.7 wt% C in run #A1.

115 In runs #A5 and #A6 performed at ~150 GPa, liquids were crystallized into the hcp  
 116 phase upon quenching temperature to 300 K, which included minimal amounts of H  
 117 (<0.04 wt%) (Table S1). The EPMA data showed 10.4–13.0 wt% O and 0.9–3.7 wt% C  
 118 in quenched liquids, indicating they were Fe-O±C liquids. The liquid  
 119 Fe-13.0wt%O-0.9wt%C obtained in run #A5 coexisted with solid Fe, giving the lower  
 120 bound for O concentration in the Fe-FeO binary eutectic liquid at 147 GPa (Figure S1).  
 121 It helps to constrain the change in the eutectic liquid composition in the Fe-FeO system  
 122 with increasing pressure, which was previously estimated only with a single datum  
 123 point above 50 GPa in Oka et al. (2019).

## 124 2.2. Melting in Fe-H

125 Melting experiment was carried out also on the Fe-FeH binary system at 45 GPa (run  
 126 #B, Table S2) (see Experimental Methods in the Supporting Information).  
 127 Non-stoichiometric hcp  $\text{FeH}_{0.30}$  was synthesized by heating to <1000 K at 10 GPa  
 128 (Figure 1b). The excess molecular  $\text{H}_2$  in a sample chamber was lost to a neighboring  
 129 rhenium gasket to form Re-H alloy upon compression to 23 GPa at room temperature  
 130 (Scheler et al., 2011). After further compression, we melted this sample by heating to  
 131 1900 K for 80 sec at 45 GPa. The hcp 002, 101, and 102 peaks were observed in the  
 132 XRD pattern during heating, and the presence of liquid was confirmed when quenching  
 133 temperature by the appearance of the new hcp 100 and 110 peaks and of the hcp 002,  
 134 101 and 102 additional spot peaks in the 2D XRD image. We found the H content in

liquid to be  $\text{FeH}_{0.26}$  from these hcp peaks that emerged upon temperature quench (Figure 1b). The solid phase that coexisted with liquid was hcp  $\text{FeH}_{0.20}$  based on spot peaks that were present during heating. These results show the solid/liquid partition coefficient of H,  $D_{\text{H}} = 0.77$  on weight basis. Carbon contamination should have been minor in this experiment because the sample was heated only to 1900 K, much less than in other melting experiments on Fe-O-H.

### 3. Discussion

#### 3.1. Eutectic Melting in Fe-FeH and the Liquidus Phase Relations in Fe-O-H

The high-pressure phase relations in the Fe-H system have been speculated by Fukai (1992), suggesting 1) continuous solid solution between Fe and FeH and 2) an eutectic composition with  $\text{FeH}_x$  ( $x > 1$ ). The multi-anvil experiments by Imai (2013) and Shibazaki et al. (2014) reported the melting and subsolidus phase relations in the Fe-FeH( $\pm$ Ni) system to 15–20 GPa, which are consistent with the phase diagram supposed by Fukai (1992). In contrast, in the present experiment (run #B), melting occurred by heating to 1900 K at 45 GPa, lower than the melting point of 2100 K for stoichiometric FeH (Tagawa, Helffrich et al., 2022) (note that if Fe and FeH form continuous solid solution, the solidus temperature of  $\text{FeH}_{0.20-0.30}$  should be closer to the melting point of pure Fe than that of FeH). In addition, we found  $\text{FeH}_{0.20}$  with the hcp structure at this  $P$ - $T$  condition, which is different from the fcc structure for FeH (Kato et al., 2020; Thompson et al., 2018; Tagawa, Gomi et al., 2022). It indicates a gap in solid solution between Fe and FeH. These suggest eutectic (not peritectic) melting between the H-poor hcp and H-rich fcc phases in the Fe-FeH system. The presence of the gap in solid solution between hcp Fe and fcc FeH and the resulting eutectic melting are likely to be hold to Earth's inner core conditions (Tagawa, Helffrich et al., 2022).

The compositions of liquids obtained at  $\sim 40$  GPa in runs #A1–A4 and #B are plotted in the Fe-O-H ternary diagram (Figure 3a). The liquid compositions found in runs #A2 and #A3 should be within the liquidus field of FeO, and that in run #A4 is on the Fe + FeO cotectic line (showing the liquid compositions coexisting with both Fe and FeO). Considering the eutectic point in the Fe-FeO binary (Figure S1), the position of the Fe + FeO cotectic line is tightly constrained. Figure 3a illustrates the liquidus phase relations in the Fe-O-H system. The liquidus field of FeO extends close to the Fe-FeH side, and the ternary peritectic (unlikely eutectic) point should be located at the H-rich portion of the phase diagram. It suggests the presence of the Fe-FeH binary eutectic point at  $\text{FeH}_{0.42}$  (Fe + 0.75 wt% H). It includes more hydrogen than the liquid  $\text{FeH}_{0.26}$  obtained

169 at 1900 K in run #B, suggesting that the Fe-FeH binary eutectic temperature is  
170 somewhat lower than 1900 K at 45 GPa.

171 The Fe-FeH binary eutectic liquid composition may change little with increasing  
172 pressure because the high  $P$ - $T$  experiments performed by Tagawa, Helffrich et al. (2022)  
173 showed that the temperature/pressure slope of the melting curve of stoichiometric FeH  
174 is similar to that of Fe at >40 GPa. On the other hand, the present results on the Fe-FeO  
175 system as well as earlier experiments (Seagle et al., 2008; Oka et al., 2019) and  
176 thermodynamic calculations (Komabayashi, 2014) demonstrate that O concentration in  
177 the Fe-FeO binary eutectic liquid increases to 15 wt% with increasing pressure to 330  
178 GPa. The Fe-O±C liquids obtained in runs #A5 and #A6 verify the pressure evolution of  
179 the Fe-FeO eutectic liquid composition (the composition of liquid that coexists with  
180 both Fe and FeO) in the presence of <1 wt% and ~3 wt% C, respectively (Figure S1).  
181 By taking the positions of the Fe-FeO and Fe-FeH binary eutectic points into account,  
182 we suppose the Fe-O-H ternary liquidus phase relations at the ICB pressure in Figure  
183 3b.

### 184 3.2. Possible Outer Core Liquid Composition

185 Since the outer core crystallizes the dense inner core at the ICB, the liquid composition  
186 should be within the liquidus field of Fe at 330 GPa (Figure 3b). It has been  
187 demonstrated by earlier experiments on ternary Fe alloy systems containing two light  
188 elements—Fe-Si-S (Tateno et al., 2018), Fe-S-O (Yokoo et al., 2019), Fe-Si-C  
189 (Hasegawa et al., 2021), and Fe-C-O (Sakai & Hirose, 2021)—that the ternary eutectic  
190 point is located close to the tie line between the eutectic points in relevant binary  
191 systems; in other words, the liquidus field of Fe (such as a colored area in Figure 3b) in  
192 these ternary systems can be approximated by linear interpolation between the eutectic  
193 liquid compositions in relevant binary systems. This may be extended to the  
194 Fe-O-H-Si-S system that we consider for the liquid outer core; the liquidus field of Fe  
195 could be estimated from the four relevant binary eutectic compositions at 330 GPa; Fe  
196 with 15 wt% O (Figure S1), 0.75 wt% H (Figure 3b), 8 wt% Si (Hasegawa et al., 2021),  
197 or 5 wt% S (Mori et al., 2017).

198 We estimate the possible range of the outer core liquid composition based on three  
199 independent constraints (Hirose et al., 2021). First, 1) it must account for the density  
200 and P-wave velocity observed in the outer core (Umemoto & Hirose, 2020). We first  
201 consider  $T_{\text{ICB}} = 5800$  K and 6280 K, which correspond to  $T_{360\text{GPa}} = 6000$  K and 6500 K,  
202 respectively, when assuming isentropic temperature profiles with Grüneisen parameter  $\gamma$

203 = 1.5 (Vočadlo et al., 2003); these are the conditions at which the possible inner core  
 204 composition was examined by Wang et al. (2021). Also, 2) the liquid core composition  
 205 should be within the liquidus field of Fe in the Fe-O-H-Si $\pm$ 1.7wt%S system at the ICB  
 206 pressure. Geochemical and cosmochemical estimates have consistently proposed  $\sim$ 2  
 207 wt% S in the core (Dreibus & Palme, 1996; McDonough, 2014), and here we adopt 0 or  
 208 1.7 wt% S in the outer core according to Dreibus & Palme. The colored portions in  
 209 [Figures 4a–d](#) indicate O, H, and Si concentrations in liquids Fe-Ni-O-H-Si $\pm$ 1.7wt%S  
 210 (Fe/Ni = 16 by weight), which meet these two constraints from 1) the outer core density  
 211 and velocity and 2) the liquidus phase diagram at 330 GPa.

212 In addition, 3) the third constraint is from the possible inner core solid composition that  
 213 was estimated in Fe-Si-S(-C) by Li et al. (2018) and in Fe-H-Si by Wang et al. (2021),  
 214 which explains the observed density, P-wave and S-wave velocities (Dziewonski &  
 215 Anderson, 1981; Kennett et al., 1995). Their calculations of Fe<sub>64-y</sub>Si<sub>y</sub> ( $y = 0, 4$ , and  $8$ )  
 216 and Fe<sub>60</sub>Si<sub>4</sub>H<sub>z</sub> ( $z = 1, 2, 4$ , and  $8$ ) alloys demonstrated that the density ( $\rho$ ), P-wave ( $V_P$ )  
 217 and S-wave velocities ( $V_S$ ) of Fe alloys are written as;  $\rho = -0.105y - 0.039z + 13.639(5)$ ,  
 218  $V_P = -0.001y - 0.034z + 11.559(128)$ , and  $V_S = -0.061y - 0.053z + 4.108(154)$  at  $T_{360\text{GPa}}$   
 219  $= 6500$  K, and  $\rho = -0.104y - 0.038z + 13.682(48)$ ,  $V_P = 0.001y - 0.019z + 11.619(110)$ ,  
 220 and  $V_S = -0.060y - 0.033z + 4.229(140)$  at  $T_{360\text{GPa}} = 6000$  K. The inner core may include  
 221 1.4 wt% S, which is calculated from the 1.7 wt% S in the outer core and the solid/liquid  
 222 partition coefficient of S,  $D_S = 0.8$  by weight at 330 GPa (Alfè et al., 2002; Yokoo et al.,  
 223 2019). Since Wang et al. (2021) mentioned that the effects of S and Si on the density  
 224 and sound velocities are similar to each other, we consider that the effect of S is  
 225 equivalent to that of Si. O is not partitioned into solid Fe and thus not considered in the  
 226 inner core (Alfè et al., 2002; Ozawa et al., 2010). With these relations, we obtain the  
 227 possible ranges of the inner core composition in the Fe-H-Si $\pm$ 1.4wt%S system,  
 228 depending on temperature. It gives the liquid outer core composition based on the  
 229 solid/liquid partition coefficients  $D_H = 0.89$  (from run #A-4 in this study) and  $D_{Si} = 1.0$   
 230 by weight (Alfè et al., 2002; Kuwayama & Hirose, 2004). The  $D_H = 0.77\text{--}0.89$  obtained  
 231 in this study at 39–45 GPa is slightly higher than 0.72–0.74 reported by earlier  
 232 multi-anvil experiments at 15–20 GPa (Imai, 2013). The compositional ranges of liquids  
 233 in chemical equilibrium with the possible inner core solid are illustrated by areas  
 234 enclosed by black lines in [Figures 4a–d](#).

235 The overlap between the colored area (constrained by the outer core density and  
 236 velocity and by the liquidus field of Fe) and the black enclosed area (constrained by the  
 237 inner core density and velocities) in [Figures 4a–d](#) indicates the possible compositional

range for the liquid outer core. It becomes smaller with 1.7 and 1.4 wt% S respectively in the outer and inner core than in S-free cases. Possible liquid compositions are not found in the Fe-O-H-Si-1.7wt%S system when  $T_{\text{ICB}} = 5800$  K ( $T_{360\text{GPa}} = 6000$  K) (Figure 4c). If  $T_{\text{ICB}} = 6280$  K ( $T_{360\text{GPa}} = 6500$  K), the outer core may include 2.9–5.2 wt% O, 0.03–0.32 wt% H, and 0–3.4 wt% Si and in addition to 1.7 wt% S. Furthermore, with  $T_{\text{ICB}}$  of  $\sim 6000$  K, the H-rich liquid core with  $\sim 0.3$  wt% H and  $\sim 3$  wt% O in addition to 1.7 wt% S and minimal Si satisfies the seismological and liquidus-phase constraints considered here (Figures 4c, d). It suggests that  $\sim 1$  wt%  $\text{H}_2\text{O}$  in the bulk Earth, corresponding to about 50 ocean mass of water, is sequestered in the core (Tagawa et al., 2021; Li et al., 2020; Yuan & Steinle-Neumann, 2020). It is supported by recent planet formation theories that a large amount of water was brought to the Earth during its accretion (Raymond et al., 2007; Walsh et al., 2011). The range of the O content is still consistent with the recent estimate of  $<3.8$  wt% O in the outer core to explain the density jump across the ICB, which was based on the experimentally-determined equations of state of liquid and solid Fe (Kuwayama et al., 2020; Dewaele et al., 2006). Indeed,  $\sim 0.2$  wt.% C could be also present in the core (Fischer et al., 2020), but it hardly decreases the present estimates of the core inventories of other light elements.

The  $T_{\text{ICB}} = 6000$ – $6280$  K, however, corresponds to about  $4400$ – $4600$  K at the CMB when  $\gamma = 1.5$  (Vočadlo et al., 2003). It is much higher than the solidus temperatures of the pyrolitic and chondritic mantle materials at 135 GPa (Fiquet et al., 2010; Andrault et al., 2011; Nomura et al., 2014; Kim et al., 2020). It could therefore lead to extensive melting in the lowermost mantle at present, but seismology reveals the presence of partial melts only locally at ultralow velocity zones above the CMB. It is necessary to better constrain the core temperature to further explore the possible core composition (Hirose et al., 2021).

#### 4. Conclusions

We obtained the liquidus phase relations in the Fe-O-H ternary and Fe-H binary systems at  $\sim 40$  GPa. While earlier studies reported that Fe and stoichiometric FeH form continuous solid solution below  $\sim 20$  GPa (Sakamaki et al., 2009; Imai, 2013; Shibazaki et al., 2014), our results at  $\sim 40$  GPa indicate eutectic melting between Fe and FeH from the observations that 1)  $\text{FeH}_{0.20}$  was formed with the hcp structure, different from fcc for stoichiometric FeH, suggesting a miscibility gap between the H-poor hcp and H-rich fcc phases in Fe-FeH, 2) the liquidus temperature of  $\text{FeH}_{0.26}$  is 1900 K, lower than the melting temperature of FeH (Tagawa, Helffrich et al., 2022), and 3) melting phase



relations in the Fe-O-H ternary require the eutectic point in the Fe-FeH system. The Fe-FeH eutectic liquid composition is found to be FeH<sub>0.42</sub> (Fe + 0.75 wt% H) at this pressure. The pressure effect on the Fe-FeH eutectic liquid composition is likely to be small because the temperature/pressure slope of the melting curve of FeH is comparable to that of Fe (Tagawa, Helffrich et al., 2022). On the other hand, the present study combined with earlier experiments demonstrates that O concentration in the Fe-FeO binary eutectic liquid increases to 15 wt% at 330 GPa. These allow us to extrapolate the liquidus phase relations in the Fe-O-H system to 330 GPa.

We estimated the possible range of the outer core liquid composition to be Fe + 2.9–5.2 wt% O + 0.03–0.32 wt% H + 0–3.4 wt% Si + 1.7 wt% S when  $T_{\text{ICB}} = 6280$  K, based on constraints from 1) the liquidus field of Fe (in order to crystallize the dense inner core), 2) the outer core density and velocity, and 3) the inner core density and velocities (the outer core composition is calculated from that of the inner core using the solid-liquid partition coefficients of light elements at ICB conditions). If  $T_{\text{ICB}}$  is  $\sim 6000$  K, the outer core is enriched in both O and H, supporting recent arguments on the delivery of a large amount of water to the growing Earth and its incorporation into core metals for the most part.

#### Data Availability Statement

Datasets for this research are found in [Tables S1](#), [S2](#), and [Dataset S1](#) available online (from <https://doi.org/10.5281/zenodo.6508681>).

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497 **Figure 1.** XRD patterns collected in (a) run #A4 for the Fe-O-H sample and (b) run #B  
 498 for the Fe-H sample. Hcp, hcp FeH<sub>x</sub>; fcc, fcc FeH<sub>x</sub>; dhcp, FeH; rB1, rhombohedral B1  
 499 FeO; cor, corundum (pressure medium); Re, rhenium (gasket). The bottom profile in (b)  
 500 shows only peaks that appeared upon quenching temperature, which was used to  
 501 calculate hydrogen concentration in liquid in run #B.

502 **Figure 2.** Sample cross sections and temperature profile in run #A4. Scanning ion  
 503 microscope image (a) and X-ray map of oxygen (b) show that liquid coexisted with  
 504 solid Fe and FeO. Crystals were formed from liquid upon quenching temperature, and  
 505 their unit-cell volumes were used to calculate hydrogen concentration in liquid.  
 506 Temperature at the liquid/solid boundary is obtained by a combination of these images  
 507 and a temperature profile (c).

508 **Figure 3.** Liquidus phase diagrams of the Fe-FeO-FeH ternary system. Green lines are  
 509 cotectic lines that separate the liquidus fields of Fe, FeO, and FeH. In (a) for ~40 GPa,  
 510 diamond, circles, and square show the compositions of liquids coexisting with Fe, FeO  
 511 and Fe + FeO, respectively (filled and open circle/square symbols indicate 0.1–0.4 wt%  
 512 and 1.7 wt% C in liquids, respectively). The Fe-FeO eutectic liquid composition is  
 513 given by a red triangle from [Figure S1](#). The Fe-FeH eutectic liquid composition may be  
 514 FeH<sub>0.42</sub> (Fe + 0.75 wt% H). (b) The liquidus phase relations extrapolated to 330 GPa  
 515 considering the Fe-FeO (orange triangle, [Figure S1](#)) and Fe-FeH binary eutectic liquid  
 516 compositions (the latter could exhibit little pressure dependence). The liquidus field of  
 517 Fe at ICB conditions is illustrated by a blue area.

518 **Figure 4.** Possible ranges of the outer core liquid composition in the  
 519 Fe-O-Si-H±1.7wt%S system. Colored areas show liquid compositions that are  
 520 compatible with the observed outer core density and P-wave velocity (Umemoto &  
 521 Hirose, 2020) and within the liquidus field of Fe. The areas enclosed by black lines  
 522 indicate liquid compositions that are in equilibrium with the possible compositions of  
 523 the inner core solid that explain observed density, P- and S-wave velocities (Li et al.,  
 524 2018; Wang et al., 2021). If  $T_{\text{CMB}}$  is ~6000 K, the outer core may include ~3 wt% O and  
 525 ~0.3 wt% H as important light elements.



Figure 1.

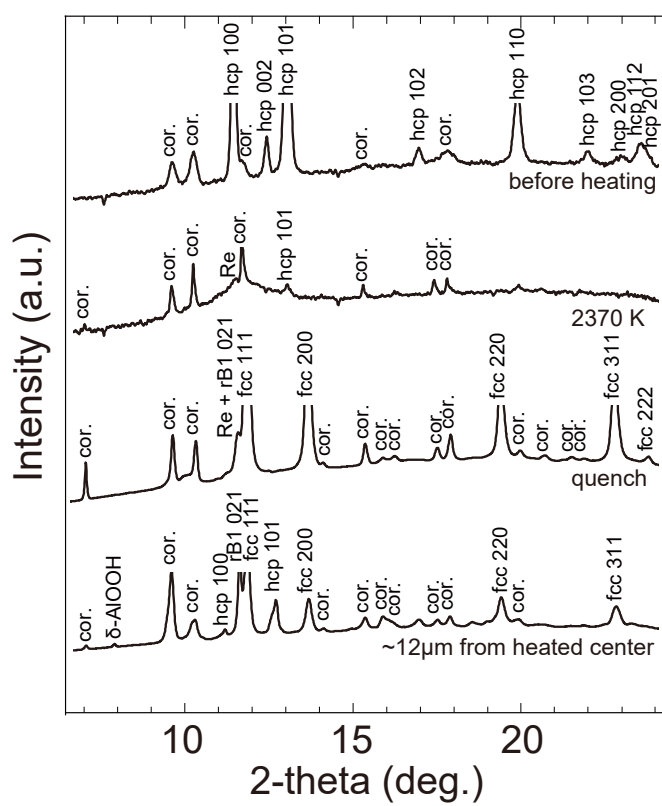
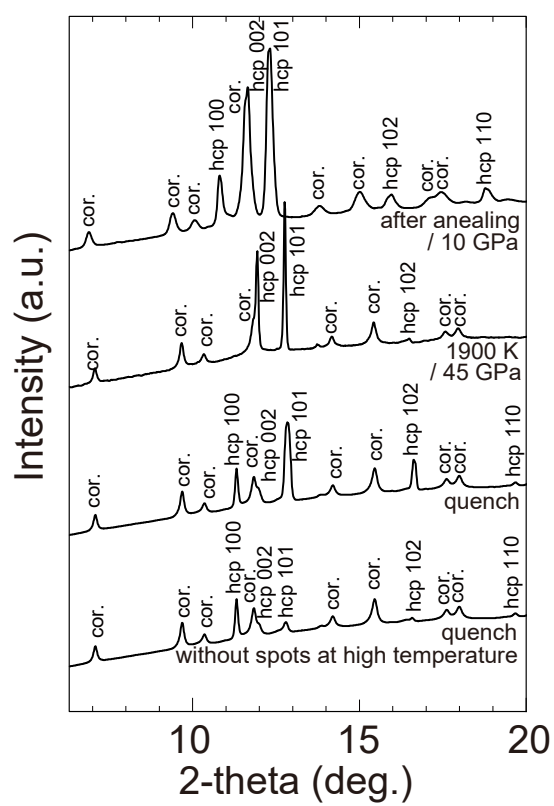
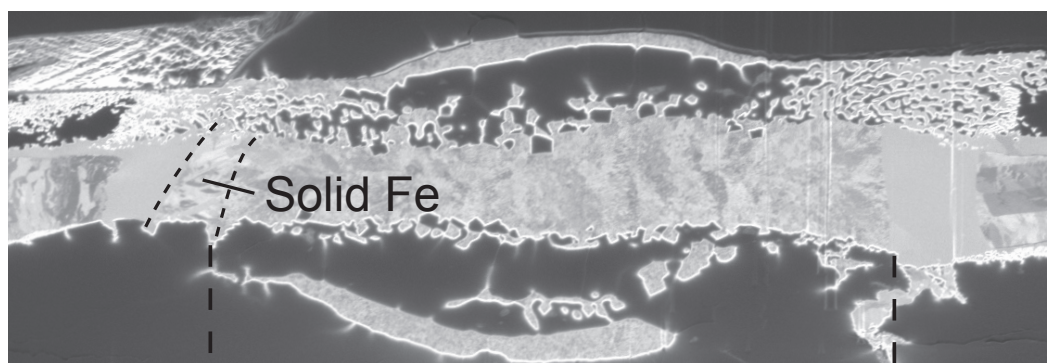
**a****b**

Figure 2.

**a**

10 $\mu$ m

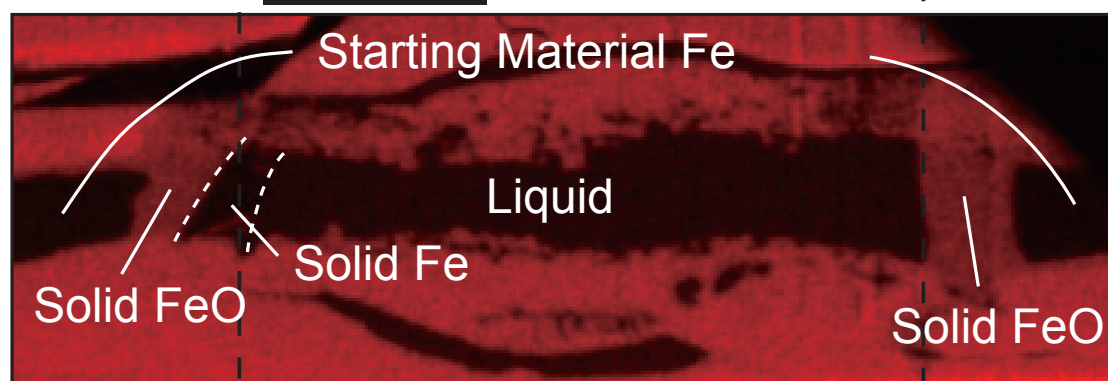
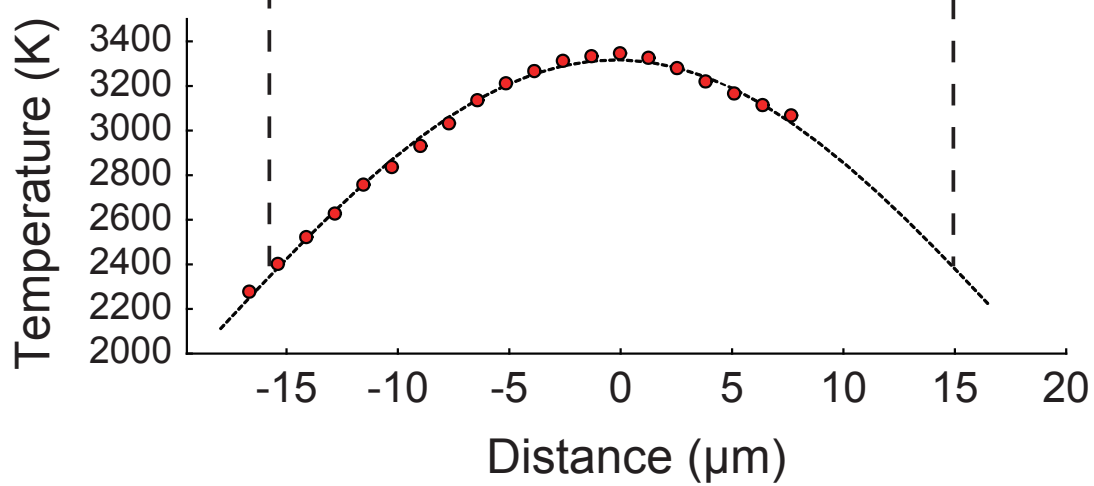
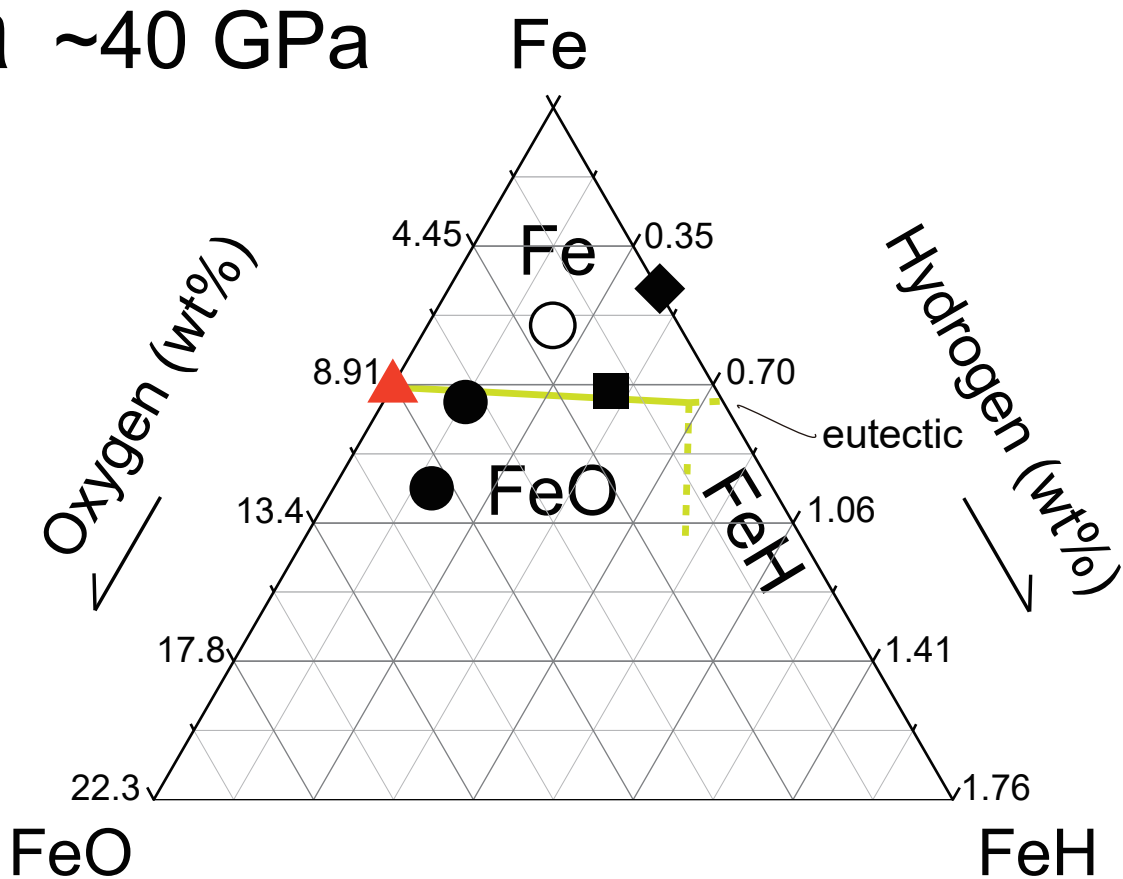
**b****c**

Figure 3.

**a** ~40 GPa



**b** ~330 GPa

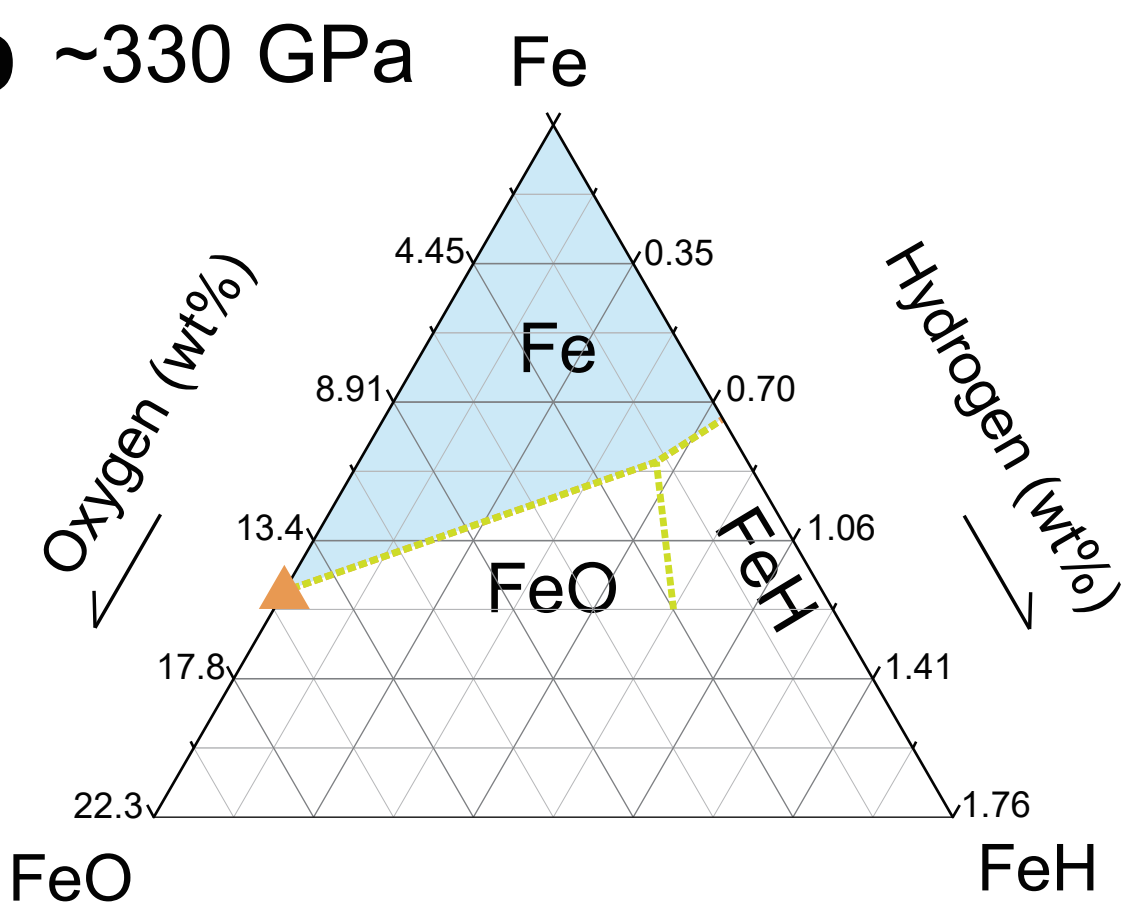


Figure 4.

